

# Measurements of Residual Stress and Microstructural Evolution in Electron Beam Welded Ti-6Al-4V Using Multiple Techniques

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## Abstract

The evolution of residual stress and microstructure has been investigated in electron-beam welded Ti-6Al-4V alloy rings in order to develop an understanding of how the distribution of through-thickness residual stress correlates with microstructural evolution. A multiple technique approach to residual stress measurement was employed using a combination of different measurement techniques including X-ray diffraction (XRD), hole drilling method based on electronic speckle pattern interferometry (ESPI), and the contour method. It was found that there is a strong correlation between the change in residual stress and alpha phase morphology across the weld. The fusion zone exhibited highly tensile residual stress which was typified by an entirely acicular  $\alpha'$  microstructure formed by a displacive transformation within prior  $\beta$  grains on cooling. The tensile residual stress in the centre of the weld reduced towards the heat affected zone, transitioning to a compressive residual stress upon increasing distance from the weld centre. The transition from tensile to compressive residual stress correlates with a significant decrease in the volume fraction of  $\alpha'$  and an increase in the bimodal morphology of equiaxed primary alpha in a diffusional transformed beta matrix leading to elongated alpha in the base material.

## Keywords

X-Ray Diffraction

Weld

Residual Stress

Contour Method

Titanium Alloys

Phase Transformation

## 1. Introduction

Titanium alloys are often selected as structural components for use in a number of applications due to their favourable strength-to-weight ratio, corrosion resistance and temperature resistance [1]. The aerospace industry accounts for majority significant amount of global titanium consumption (i.e. over 50%) and is commonly used in the airframe, landing gear and gas turbine engines [2]. The sixty-year-old  $\alpha + \beta$  type alloy Ti-6Al-4V is the most widely used titanium alloy, acting as an integral constituent of many aerospace components.

Joining technologies are essential to the manufacture of gas turbine engine parts; therefore ensuring structural integrity is pivotal to the endurance of critical components in service. Electron-beam welding is one of the most popular joining techniques in the manufacture of gas turbine engines. Electron-beam welding produces three distinct zones in the welded material including the fusion zone (FZ) which undergoes melting and re-solidification, the heat affected zone (HAZ) which is subjected to a solid state thermal cycle, and the base metal (BM) which is largely unaffected by the welding process [3]. The application of heat treatment can potentially lead to tailored mechanical properties in Ti-6Al-4V, however, thermal gradients during welding typically result in undesirable changes to the original microstructure, causing deviations from bulk mechanical properties [4]. This presents a number of challenges, most notably in the formation of undesirable residual stress and microstructure [5].

Most manufacturing processes such as forging, welding, machining and quenching are associated with the generation of residual stress [6]. Residual stresses can be thought of as internal stresses, required to maintain equilibrium, which remain in a part when it is no longer subjected to any external applied loads [7]. These residual stresses can be induced by heterogeneous plastic deformation, differential thermal fluctuations or due to phase transformation and precipitation [5]. Thus, residual stresses are often generated due to welding operations, as at least one of these mechanisms occurs during welding [8-10]. The inclusion of undesirable residual stress (i.e. tensile) can be detrimental to the life of a component [11, 12]. A recent review by Fairfax and Steinzig [13], which analysed 147 residual stress related incidents within the ASM Failure Analysis Database™, found that over 37% of incidents were directly attributable to joining-related failures. Additionally, residual stresses may lead to distortion during the manufacturing process of machining due to material removal which leads to the redistribution of residual stresses [14]. Determining the nature and magnitude of such stresses is therefore of critical importance to the manufacture and performance of welded Ti-6Al-4V components.

A limited number of investigations have been carried out which characterise the microstructure and residual stress distribution of welded Ti-6Al-4V, with little work attempting to interrelate the two [15-19]. It has been reported that the microstructure of electron-beam welded Ti-6Al-4V in the FZ, HAZ and BM is primarily composed of the following phases: martensite ( $\alpha'$ ),  $\alpha' + \alpha + \beta$ , and  $\alpha + \beta$ , respectively, with the volume fraction of  $\alpha'$  decreasing linearly across the HAZ from the FZ to the BM [17, 18]. Typically, fusion welded joints exhibit a peak tensile residual stress within the FZ, which tends to decrease throughout the HAZ, leading to balancing compressive zones in either side of the weld [20]. Further from the weld, the residual stress becomes a function of both the original residual stress existing in the BM prior to welding and those induced by the weld itself [3].

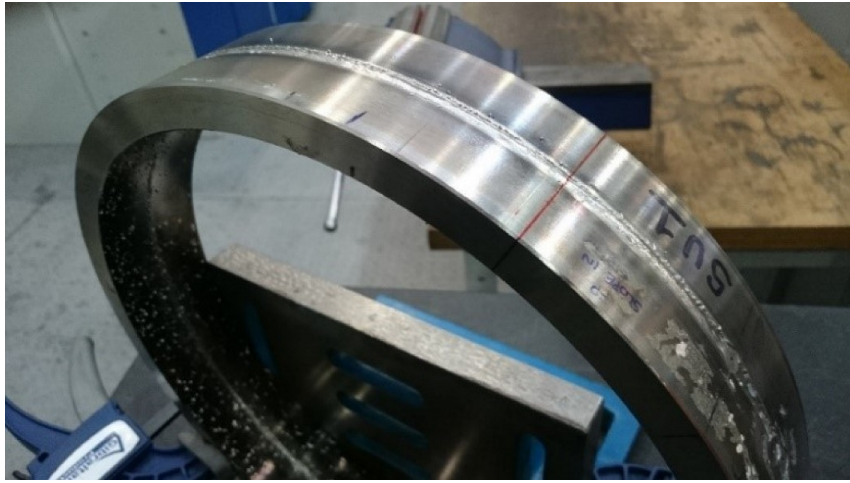
Mehdi *et al.* [16] identified and assessed both the microstructure and residual stress of Ti-6Al-4V tungsten inert gas (TIG) fusion welds. XRD was used to determine that the surface residual stress distribution is related to the prior  $\beta$  grain size in the FZ and volume fraction of different phases across the weld region. However, the use of XRD for residual stress analysis produces a collection of single point surface data, which does not allow for a through-thickness understanding of the residual stress magnitudes and distributions in a component. On the other hand, the contour method can be applied to produce a 2-Dimensional (2D) map of out-of-plane residual stress [21]. The method has also been demonstrated on Ti-6Al-4V extruded profiles [22]. Nevertheless, it has been noted that limitations in Electrical Discharge Machining (EDM) cutting quality can lead to high uncertainties within 0.5 mm of the contour sample edge [23]. Special care can be taken to mitigate these uncertainties, requiring verification by means of additional measurements using techniques such as hole-drilling and XRD [23]. Xie *et al.* employed the contour method alongside XRD in order to successfully verify the near-surface accuracy of the contour method when applied to 50 mm electron-beam welded Ti-6Al-4V plates [15]. Thus, established methods with low depth penetration such as XRD and hole-drilling techniques can be used to compliment and verify the near-surface results of the contour method. This can lead to a comprehensive understanding of through thickness residual stresses in a part which can then be interrelated to microstructural observations.

Despite the above, little work has been conducted regarding the application of multiple residual stress measurement techniques upon electron-beam welded Ti-6Al-4V to develop an understanding of how the distribution of through-thickness residual stress correlates with microstructural evolution. This paper aims to present a clear understanding of the magnitude and distribution of welding induced residual stress and microstructure evolution in electron-beam welded Ti-6Al-4V. A focus on residual stress measurement was maintained throughout this work, with the resulting findings interrelated to the microstructural evolution and microhardness profile for two weld cases.

## **2. Material and Methods**

### ***2.1 Material and Processing***

A total of two circumferentially electron-beam welded rings made from Ti-6Al-4V, labelled Ring 1 (shown in Fig. 1) and Ring 2, were selected for these investigations. The nominal chemical composition of the material used in these rings is outlined in Table 1. The two rings had dissimilar part geometries and were subjected to differing welding conditions, which allowed for the identification of common trends in residual stress and microstructural evolution, and therefore aided the transferability of the findings as well as helping to identify differences due to weld parameters. Weld parameters and ring dimensions are given in Table 2 and Fig. 2. The microstructure of the rings is largely elongated primary alpha in a transformed beta matrix. This suggests that the hot ring rolling process was conducted at subtransus temperatures.



**Fig. 1.** Photograph of Ti-6Al-4V welded sample Ring 1.

**Table 1.** Nominal chemical composition for Ti-6Al-4V alloy used in study (wt%).

V	Fe	Al	O	C	N	Ti
4	0.25	6	0.13	0.08	0.03	Bal.

**Table 2.** Weld parameters and ring dimensions (Note: The weld of Ring 1 was performed in one single pass, while Ring 2 was subjected to additional cosmetic pass, resulting in two complete passes).

Ring ID	Weld conditions				Geometry		
	Weld speed (mm/s)	Axial clamping pressure (bar)	Weld power (W)	Cosmetic pass	Outer diameter (mm)	Height (mm)	Mean thickness (mm)
1	30	210	14,250	No	353	60	16.8
2	20	'low pressure'	9,000	Yes	353	70	16.4

## 2.2 Microstructure Characterisation

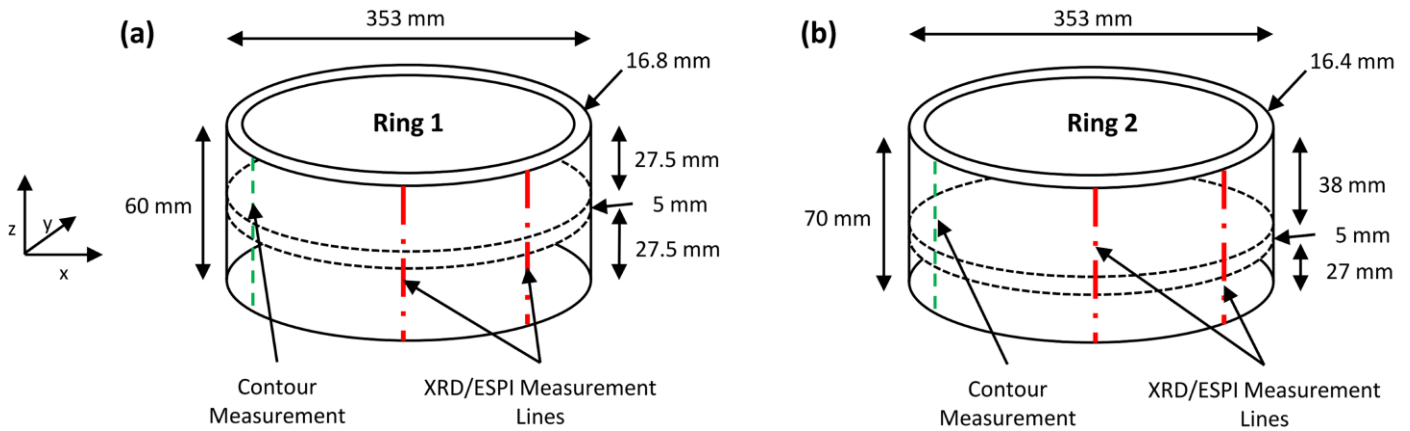
In an attempt to understand the interrelationship between microstructural evolution and the resulting residual stress distribution of electron-beam welded Ti-6Al-4V, samples were prepared for light microscopy. Specimens were sectioned radially from both rings to reveal the microstructure in the FZ, HAZ and BM close to the weld regions. Subsequent metallographic preparation was performed in line with standard techniques including the use of Kroll's Reagent (2% HF, 6% HNO<sub>3</sub>, 92% H<sub>2</sub>O) to reveal the alpha and beta phase morphology, following grinding and polishing to mirror finished condition using a Nikon Eclipse Iv1 150, and analysed using Buehler Omnimet.

### 2.3 Microhardness Measurement

In order to provide additional evidence of microstructural evolution in the welded rings, following the microstructural characterisation, a Struers Durascan G5 microhardness testing machine was used to measure a hardness map on the sample cross section; including the BM and different zones of the weld. These measurements were acquired to ASTM E-384-16 standard using a diamond shaped Vickers indenter by applying a 0.5 N load with a 15 s dwell time. Indentations were made on a rectangular grid covering the weld cross section, with 1 mm spacing in both radial direction and axial direction between indentations, and a 1 mm distance between the closest indentations and edge of the sample. Indentations were analysed with a 40 × objective lens, zoom 1, to produce a micro-hardness map.

### 2.4 Residual Stress Measurement

Residual stress analysis was carried out on each welded ring using a combination of (1) XRD, (2) hole-drilling based on ESPI, and (3) the contour method. For each welded ring, one radial cross-section was selected for the contour method, and two measurement lines were identified for XRD and ESPI stress measurements. These are shown schematically in Fig. 2. This allowed for the investigation of the axisymmetric of hoop stresses throughout the rings and, if verified, would provide additional comparison with the contour method results. A number of measurement points were selected in the FZ, HAZ and BM along these surface profiles to provide an understanding of the surface and near-surface residual stress distribution in both axial and hoop directions, and allow for later comparison with the results of the contour method.



**Fig. 2.** Schematic illustration of the welded rings with position of the welds and the dimensions of the BM material, (a) Ring 1, (b) Ring 2. The location of contour method residual stress measurements are highlighted in green, and the surface profiles for the XRD and ESPI based hole-drilling measurements are highlighted in red.

#### 2.4.1 X-ray Diffraction

The XRD technique is commonly employed to determine surface residual stresses [24]. XRD was carried out using an automated PROTO LXR X-Ray diffractometer system and the  $\sin^2\psi$  method [25]. A copper anode target was selected to obtain diffraction peaks from the {211} crystallographic planes of alpha phase. The stresses were then calculated from the strains of the {211} Bragg reflection at 140° Bragg angle, assuming

Young's modulus of 118 GPa and Poisson's ratio of 0.342. At least nine measurements were performed on the identified surface profiles for each ring (see Fig. 2) in both hoop and axial directions, at points with higher density in the weld regions and with increased point spacing towards the BM material, encompassing the weld and BM on either sides of the weld. For each point in both directions, 11 measurements with a 3 s exposure time for each measurement were performed. These were sufficient to obtain an adequate number of counts to construct strong peaks used for stress analyses. A round collimator with 1 mm radius and ten  $\psi$ -offset angles in the range of maximum  $\pm 30^\circ$  were employed. Separate gain profiles were gathered for the BM, HAZ and FZ to account for variations in Bremsstrahlung background radiation due to the differing fractions of constituent phases present throughout the material.

#### 2.4.2 Hole-drilling with ESPI

Hole-drilling based on ESPI was used to provide near surface measurements of hoop and axial residual stress components at a limited range of successive depths from the surface. This was conducted using a Stresstech Prism® system; which consisted of a high speed drill, laser source, and a CCD camera. The laser source produces a monochromatic wavelength of 532 nm. Tungsten carbide drill bits were used when carrying out the procedure. Two drill bit diameter sizes of 1.8 mm for measurements on the BM, and 1 mm for those on the FZ were used. Holes were drilled at depth increments up to 60% of the drill diameter corresponding to 1 mm depth for the BM, and 0.6 mm for the FZ. For each ring, two sets of measurements were performed in the BM ( $z = +20$  mm), with a further two in the FZ ( $z = 0$  mm) on the surface profiles, at precisely the same spots as those measured by XRD, shown schematically in Fig. 2. The through depth residual stress profiles were then calculated from the optically measured surface displacement data assuming a Young's modulus of 118 GPa and a Poisson's ratio of 0.342.

#### 2.4.3 Contour Method

The contour method was carried out on both welded rings in order to provide a full 2D map of hoop residual stress distribution in the samples. This included the task of releasing bending moments, as described by Prime [26], which act as an additional source of hoop residual stress. The measurement for each ring was carried out in four main steps: (i) sample cut, (ii) measurement of cut surface profiles, (iii) data processing, and (iv) calculation of residual stress using finite element simulations; that are discussed in the following.

A total of two cuts were made by electrical discharge machine (EDM) on each samples using a 250  $\mu$ m brass wire in 'skim' cut mode in the radial direction on the radial-axial plane starting from the outer diameter. For each sample, the first cut was conducted to release the bending moments and quantify the opening caused by the partial relaxation of hoop stress induced by bending moment [26]. For this purpose, a pair of scribed lines separated by 10 mm was made along the axial direction of the sample, and then the cut was conducted between the lines. Following the first cut and unclamping the sample, the relative displacement of the scribed lines was measured. The measured displacements were used as boundary conditions for the calculation of bending moment stresses. The second cut was then conducted at an increment of  $120^\circ$  clockwise from the first cut that

was used as the basis for residual stress measurement by contour method. The sample was clamped on both sides of the cut planes appropriately during the cutting process.

Upon completion of the EDM cuts, for each sample the outline and surface topography were measured on either side of the skim cut in a temperature controlled environment using a Mitutoyo Crysta Apex C coordinate measuring machine (CMM) with a 2 mm ruby attached to a Renishaw PH10T touch probe. The CMM equipment was programmed to acquire data with a spatial resolution of 400  $\mu\text{m}$  for points in both in-plane directions.

The measured data obtained for both surfaces of each skim cut were cleaned to minimise outliers and system noise. The data from both sides of the cut were then aligned in the same coordinate system by considering the whole perimeter of each half of the cut as an alignment guide to match the whole outlines. Following the alignment process, the datasets for both sides were linearly interpolated onto a common grid, after which the two sets were averaged. The pitch of the common grid was equivalent to that of the original data in order to produce the average surface point by point. A set of knot spacing ranging from 0.4 mm to 15.4 mm was chosen and a bivariate cubic spline for each knot spacing was fit to the averaged data. A knot spacing of about 4 mm was selected for both rings as an appropriate value.

A 3D finite element (FE) model assuming isotropic elasticity with a Young's modulus of 118 GPa and a Poisson's ratio of 0.342 was used to evaluate the residual stress from the surface contour data. For this purpose, the perimeter of the cross-section for each sample was modelled using the aligned outline measured by the CMM. The 2D cross-section was then revolved circumferentially for 180° and 120° to produce 3D models, respectively for the calculations of bending moment induced stress and residual stress. Mesh seeds were attributed to each node on the outline and also along the revolved perimeter of the model that are used as guidelines for the generation of mesh. The establishment of the seeds were biased by allocating a finer pitch near the cross-section of the model representing the cut surface, and coarser pitch towards the other end of the model that is of a less importance for stress analysis. The parts were modelled in Abaqus CAE and meshed with an element spacing of 0.8 mm on the cut surface, and 0.8–10 mm with single bias away from the surface. The selected element type was C3D20R; 20 node brick quadratic elements, as used by Prime *et al.* [27].

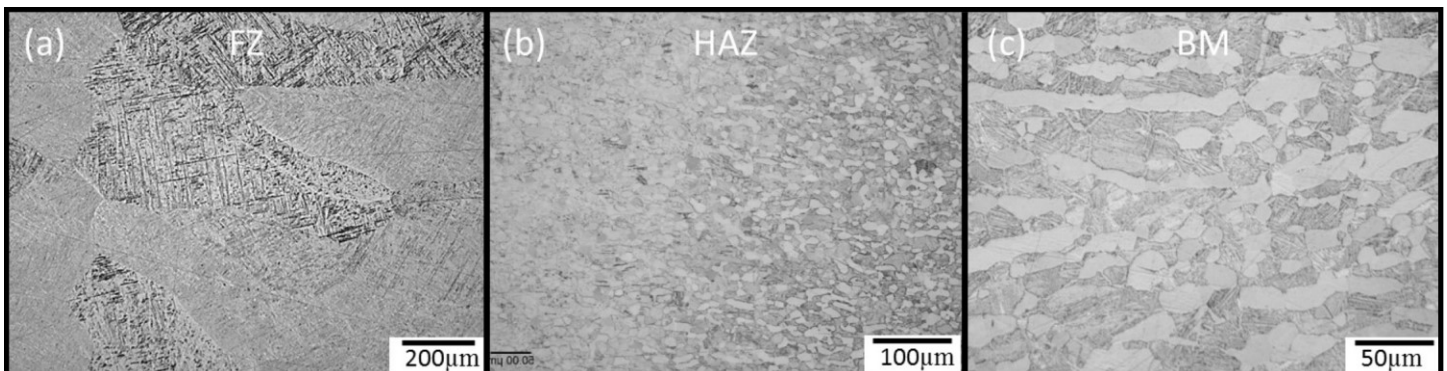
For the residual stress model, the stresses were calculated by forcing the cut surface into the opposite shape of the averaged contour data [22, 23]. For the calculation of the stresses released after the first cut (i.e. bending moment) a symmetry plane was used to constrain one surface of the model while applying bending moment on the other surface by using concentrated forces. The stresses induced by the bending moment were considered as those achieved after the deformation of the symmetry model for the same magnitude of opening displacement, with opposite sign, measured experimentally following the first cut for each sample. Both the bending moment induced stresses and the residual stresses calculated on the cut surface were integrated to achieve the final magnitude and distribution of residual stress.

### 3. Results

#### 3.1 Weld Microstructure

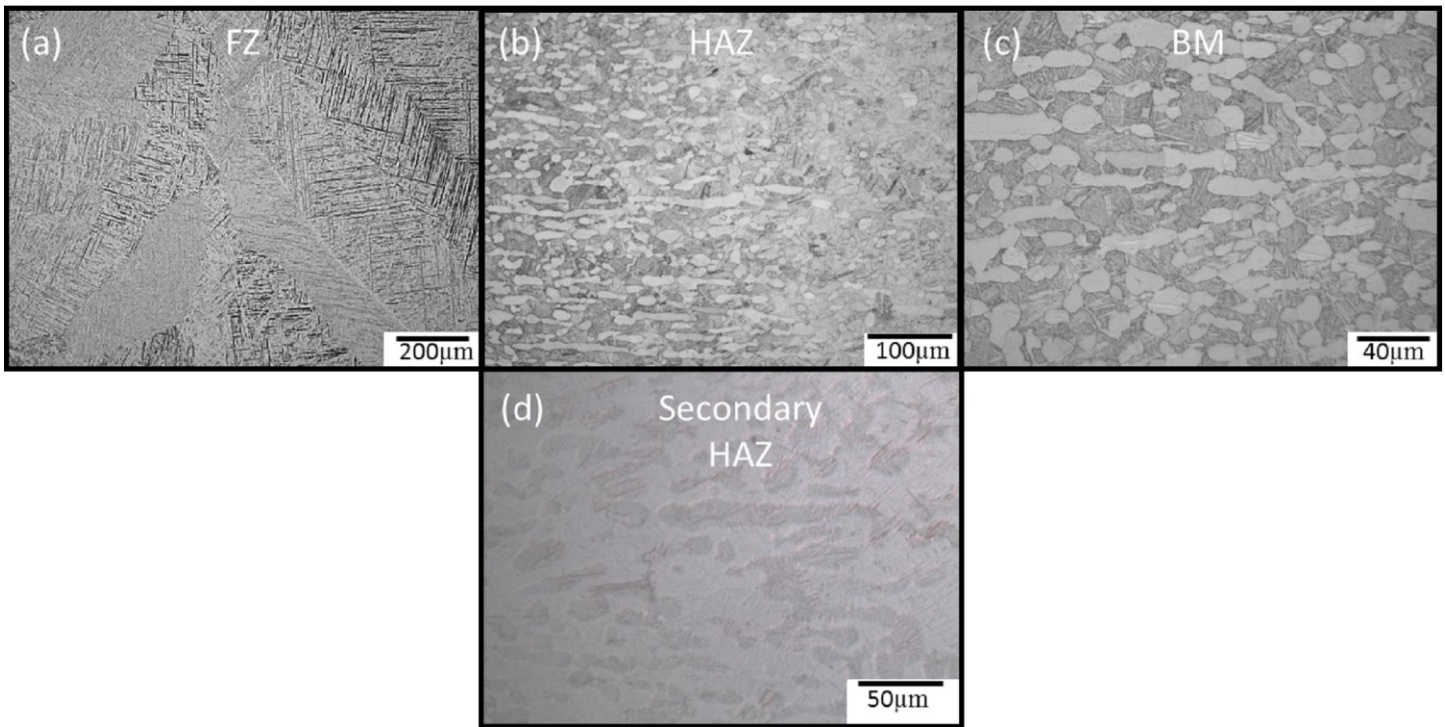
The light micrographs obtained for Ring 1 show three distinct microstructures for different regions of the weld, each characterised by varying microstructural features (Fig. 3). These regions include the FZ, HAZ and BM, the microstructures of which are dictated by the thermal history during welding. It is evident from Fig. 3a that the FZ is almost fully comprised of acicular  $\alpha'$  in prior columnar  $\beta$  grains which is a characteristic of diffusion-less shear transformation of  $\beta$  upon fast cooling from the welding temperature. There is some evidence of epitaxial growth of these columnar FZ grains originating from the HAZ. The HAZ can be characterised as a transition region with regards to microstructure and can therefore be subdivided into the near and far HAZ with respect to the FZ; where the microstructure is dependent on the local temperature in relation to the material's martensitic start temperature ( $M_s$ ). It is apparent from the micrograph presented in Fig. 3b that the near HAZ (left) primarily consists of acicular  $\alpha'$ , similar to that of the FZ yet with a distinct morphology. The far HAZ (right), however, has a microstructure more representative of the BM. From Fig. 3b there is a visible decrease in the volume fraction of  $\alpha'$  from near to far HAZ as the local temperature during welding decreases below the  $M_s$ . As volume fraction of  $\alpha'$  decreases, it is replaced by an increasing presence of fine scale globular  $\alpha$  and  $\beta$  lamellae in prior  $\beta$  grains, and equiaxed  $\alpha$  at prior grain boundary  $\alpha$  ( $\alpha_{gb}$ ). Therefore between the FZ and the HAZ there is a transition from a shear dominated to diffusion dominated phase transformation. The fine scale primary alpha becomes coarsens and becomes elongated towards the BM, and is shown in Fig. 3c.

Similarly to Ring 1, light micrographs of Ring 2 showed evident signs of microstructurally unique FZ, HAZ and BM regions, as shown in Fig. 4. However, there was an additional 'secondary HAZ' attributed to the cosmetic pass (see Fig. 4d), identified as a fourth region close to the outer diameter of the ring; replacing the original FZ in this region, see Fig. 5 for details. There is an evident macrotexture in the secondary HAZ with a directionality towards the centre of the weld. Further increases in magnification revealed faint equiaxed  $\alpha$  grains and transformed  $\beta$ , identified as 'ghost  $\alpha'$ ', shown in Fig. 4d.

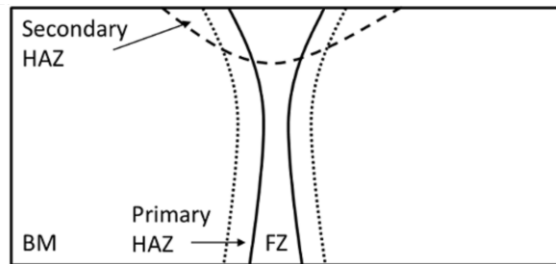


**Fig. 3.** Optical micrographs of the weld zones of Ring 1. (a) FZ, (b) HAZ, and (c) BM.





**Fig. 4.** Optical microscopy appearances of the weld zones of Ring 2. (a) BM, (b) HAZ, (c) FZ, and (d) Observed 'ghost  $\alpha$ ' in the 'secondary HAZ' due to cosmetic pass.



**Fig. 5.** Schematic representation of secondary HAZ formed within the cross section of Ring 2 induced by the additional cosmetic weld pass.

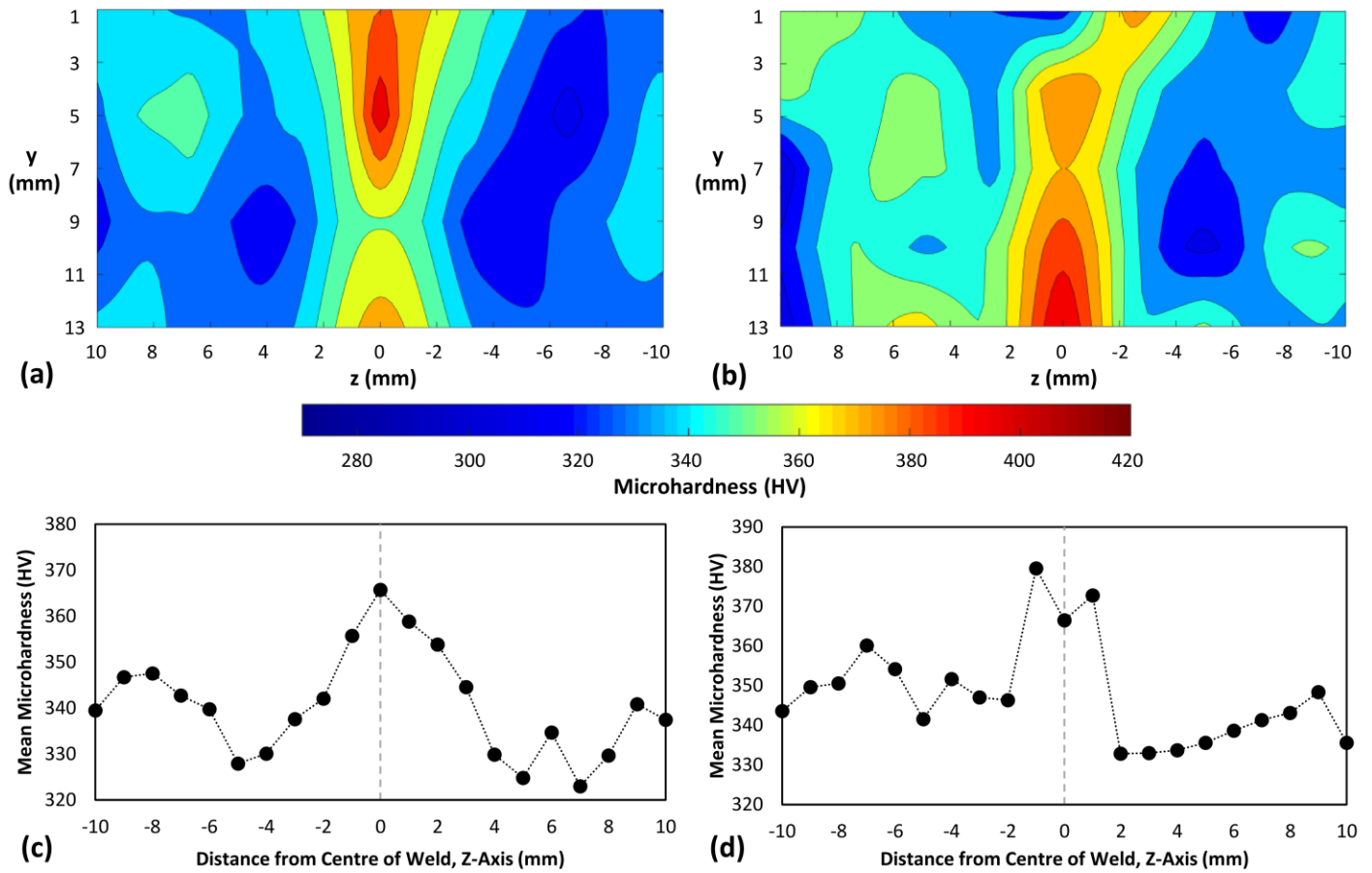
### 3.2 Weld Microhardness

The microhardness results presented in Fig. 6 provide further evidence of microstructural evolution and strength variations across the weld, extending from the BM on one side to the BM on the other side, covering all the regions. An area of 21 mm  $\times$  12 mm was considered for each ring; starting at a depth of 1 mm from the outer surface, and axially centred on the weld.

In Ring 1, the maximum microhardness reading was 416 HV, with a minimum of 273 HV; whereas values ranged from 292 to 416 HV for Ring 2. The average hardness was found to increase from  $\sim$  330 HV in the BM to  $\sim$  370 HV in the FZ of both Ring 1 and Ring 2. The obtained microhardness data for the HAZ shows a transition between the FZ and BM for both rings, correlating with the optical micrographs presented in Fig. 3 and Fig. 4.

The FZ of both rings were typified by a region of increased hardness, as shown in Fig. 6a and b. This tended to decrease throughout the HAZ towards the BM; which generally displayed a lower hardness. The overall trend in microhardness is clear, however, there were isolated patches of increased hardness in the BM, which is

noticeable in Fig. 6b. The line plots of averaged y-axis microhardness across the weld in the z-direction (Fig. 6c and d) further highlight the increased hardness of the FZ compared to the BM in both rings.

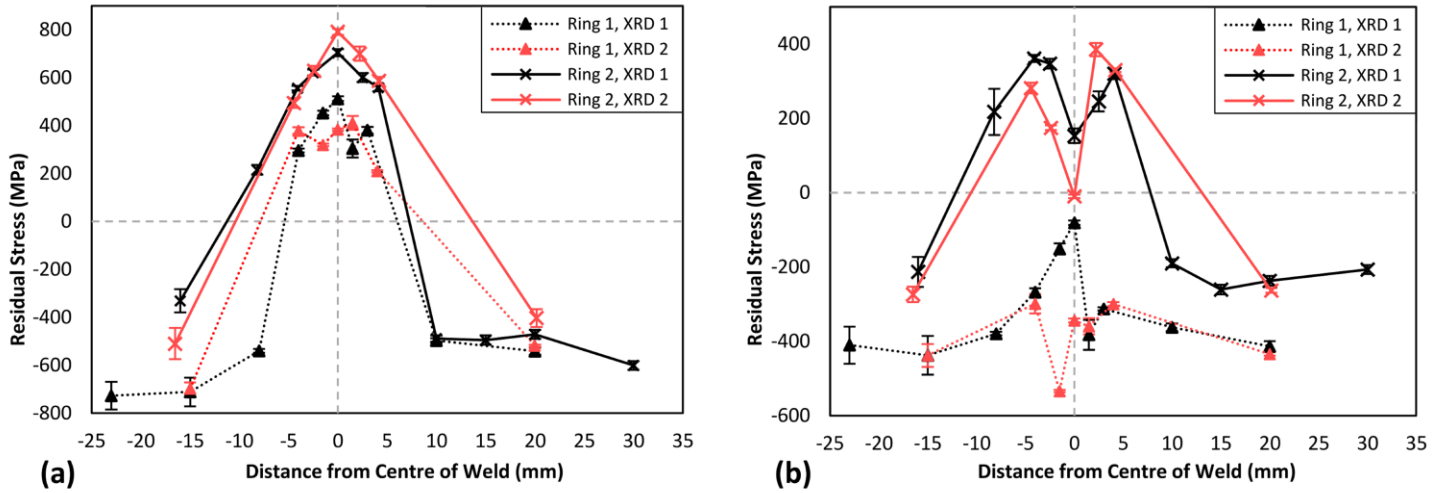


**Fig. 6.** 2-D microhardness maps across the electron-beam welded regions in (a) Ring 1, and (b) Ring 2. Accompanying line plot of mean microhardness across the weld in (c) Ring 1, and (d) Ring 2. .

### 3.3 Weld Residual Stress

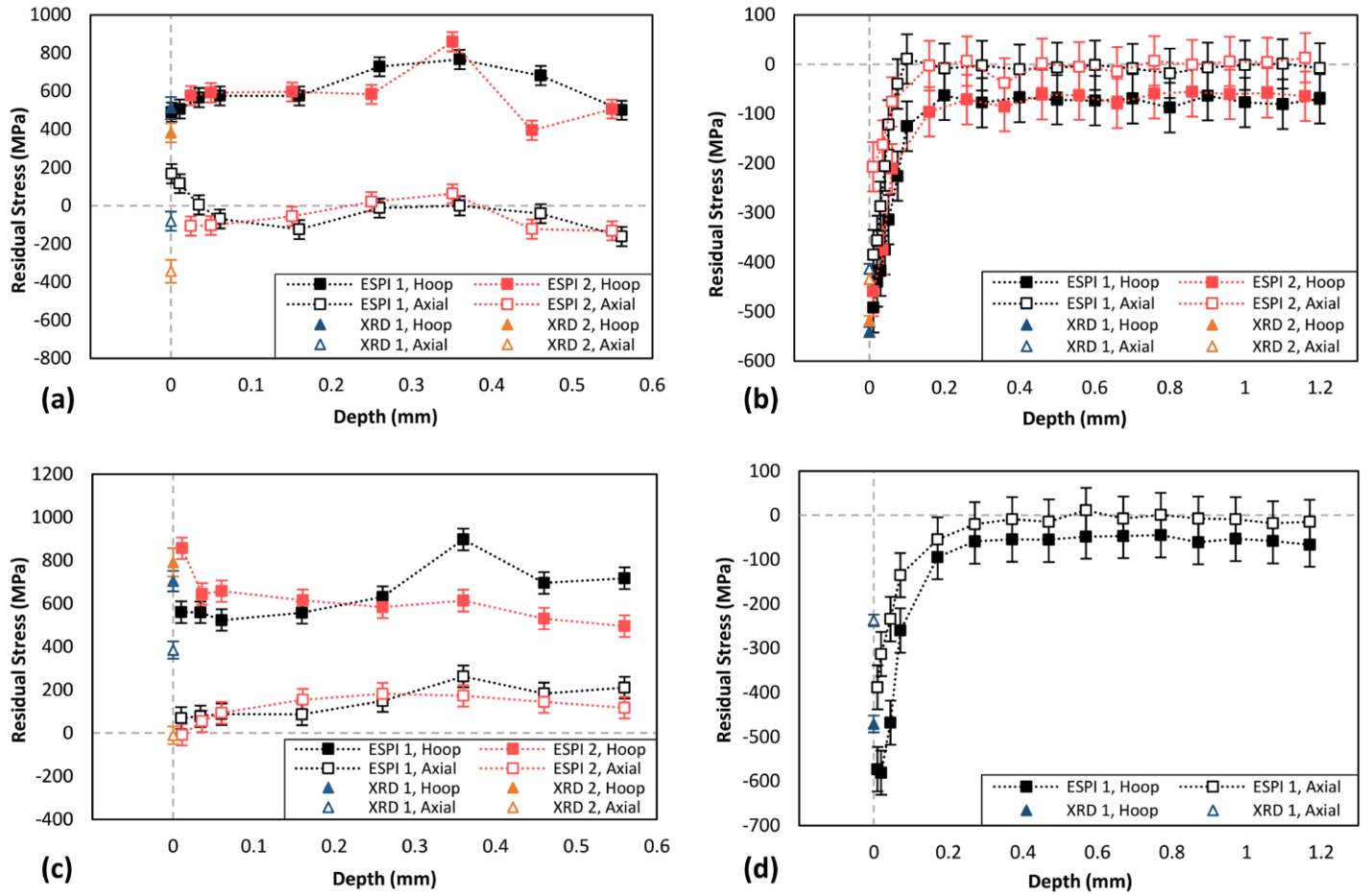
The hoop and axial surface residual stress components obtained by XRD for both samples are presented in Fig. 7a and Fig. 7b, respectively. These results display a tensile hoop stress region in the FZ and HAZ, with peak residual stresses of  $+ 513 \pm 58$  MPa for Ring 1 and  $+ 792 \pm 65$  MPa for Ring 2, in the centre of the weld. Compressive residual stresses were measured in the BM of both samples with peak values of  $- 727 \pm 10$  MPa for Ring 1 and  $- 601 \pm 18$  MPa for Ring 2. In Ring 1, all measured axial surface stresses were compressive; ranging from  $- 81 \pm 50$  MPa to  $- 534 \pm 51$  MPa. On the other hand, Ring 2 generally exhibited tensile axial residual stresses in the HAZ and FZ with a maximum of  $+ 385 \pm 34$  MPa, reducing to a more stable compressive residual stress as distance from weld centre increased. There is a good agreement between the data collected for both surface profiles (i.e. XRD 1 and XRD 2); particularly in the BM, providing evidence of an axisymmetric residual stress distribution across the sample surface. It should be noted that there were some discrepancies observed between the XRD residual stress results in the FZ, however this may be due to the residual stress gradient across the weld coupled with the  $\pm 1$  mm lateral spatial resolution of the technique [24].

Depth profiles of residual stress distribution quantified by hole-drilling based on ESPI for the FZ ( $z = 0$  mm) and BM ( $z = +20$  mm) of both samples are provided in Fig. 8, displaying the data obtained for both surface profiles (i.e. ESPI 1 and ESPI 2) in the hoop and axial directions. The corresponding surface residual stresses for these zones, as quantified by XRD, are also presented in Fig. 8 for reference. In the FZ, tensile hoop stresses were measured at all depths, with maximum values of  $+860 \pm 50$  MPa at a depth of 0.35 mm for Ring 1 (see Fig. 8a) and  $+898 \pm 50$  MPa at a depth of 0.36 mm for Ring 2 (see Fig. 8c). Axial residual stresses in the FZ of Ring 1 were found to vary between tensile and compressive in nature with very small magnitudes as depth increased (see Fig. 8a). However, axial stresses were found to be mostly tensile for the FZ of Ring 2 (see Fig. 8c), with a maximum value of  $+263 \pm 50$  MPa at a depth of 0.36 mm. Both rings showed a similar trend in BM residual stresses; with highly compressive hoop and axial stresses leading to a relatively stable value (see Fig. 8b and d). This ranged from between roughly 0 to  $-50$  MPa and  $-50$  MPa to  $-100$  MPa for axial and hoop stresses, respectively, at depths of 0.1 mm to 1.2 mm. The ESPI based hole-drilling results provide further confirmation of axisymmetric residual stress distribution; with good accordance between surface profiles upon increasing depth.



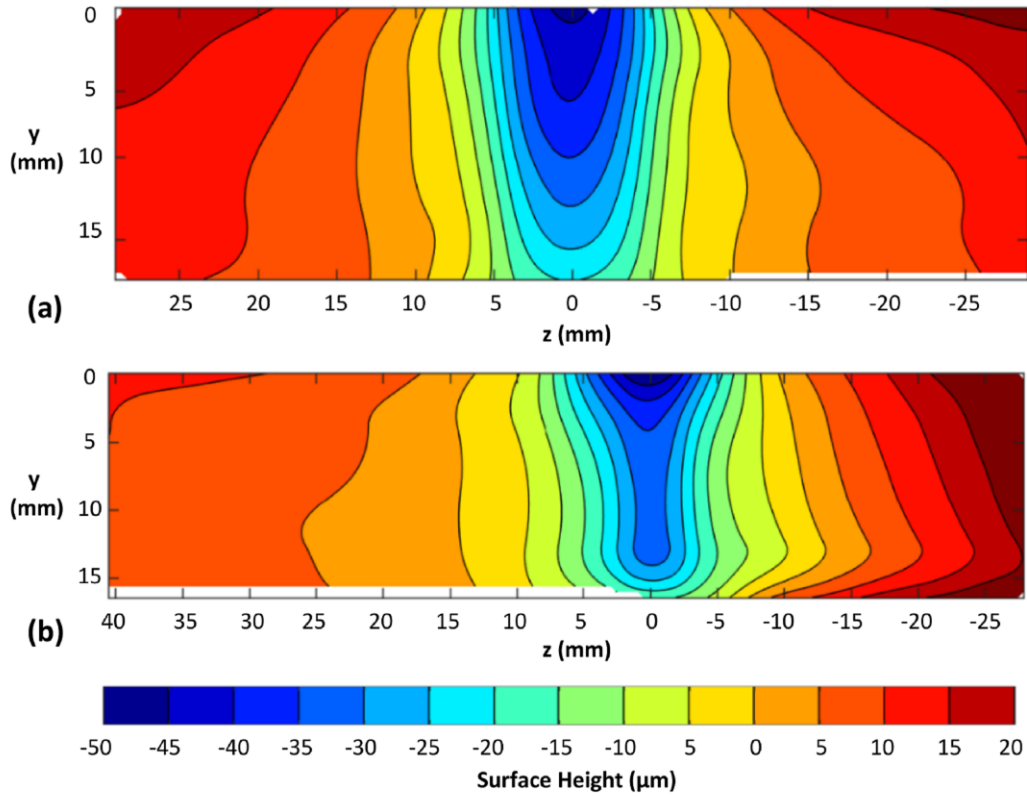
**Fig. 7.** XRD residual stress data across the weld for Ring 1 and Ring 2. (a) Hoop stress, and (b) Axial stress.

The opening distances measured between the scribed lines induced by bending moments were 2 mm and 2.2 mm for Ring 1 and Ring 2, respectively. These correspond to stresses which vary almost linearly from approximately 25 MPa on the outer surface to about 35 MPa on the inner surface for both rings. Fig. 9 shows the contour plots of the averaged surface height obtained by the CMM for both samples, following the EDM skim cut for contour residual stress measurement. The range of peak to valley of the contours was roughly 70  $\mu$ m for Ring 1 and Ring 2. The contour plots show that the surface height in both samples is low in the weld region implying that tensile residual stresses are relieved. The out-of-plane hoop residual stresses for Ring 1 and Ring 2, as evaluated by combining the bending moment stresses and the results of contour method, are provided in Fig. 10.



**Fig. 8.** Surface and near surface residual stress profiles measured by XRD and hole-drilling based on ESPI as functions of depth at different regions of Ring 1 and Ring 2 welds. (a) FZ of Ring 1 at  $z = 0$  mm, (b) BM of Ring 1 at  $z = +20$  mm, (c) FZ of Ring 2 at  $z = 0$  mm, and (d) BM of Ring 2 at  $z = +20$  mm.

From the out-of-plane hoop stress map for Ring 1 provided in Fig. 10a, it can be seen that tensile residual stresses (i.e. light blue to red) were concentrated within the FZ and HAZ. The region of highest tensile residual stress was identified in the FZ between depths of 3–9 mm on the y-axis, with the peak stress reaching + 495 MPa. The remainder of the sample surrounding the FZ and HAZ was in a state of compression, as expected. Highly compressive pockets of residual stress (i.e. dark blue) were identified at approximately  $-7.5$  mm and  $+7.5$  mm the weld centre in the z-axis, with a peak compressive stress of  $-300$  MPa exhibited. Additionally, regions of increased compressive stress were found near the outer diameter of the BM on either side of the weld. The contour method residual stress results for Ring 2 (see Fig. 10b) showed two regions of high tensile stress in the FZ; one on the outer surface, and the other between the centre and inner surface of the part. The maximum tensile stress across the cut plane exceeded that of Ring 1, reaching a peak residual stress of + 755 MPa near the outer surface of the FZ.

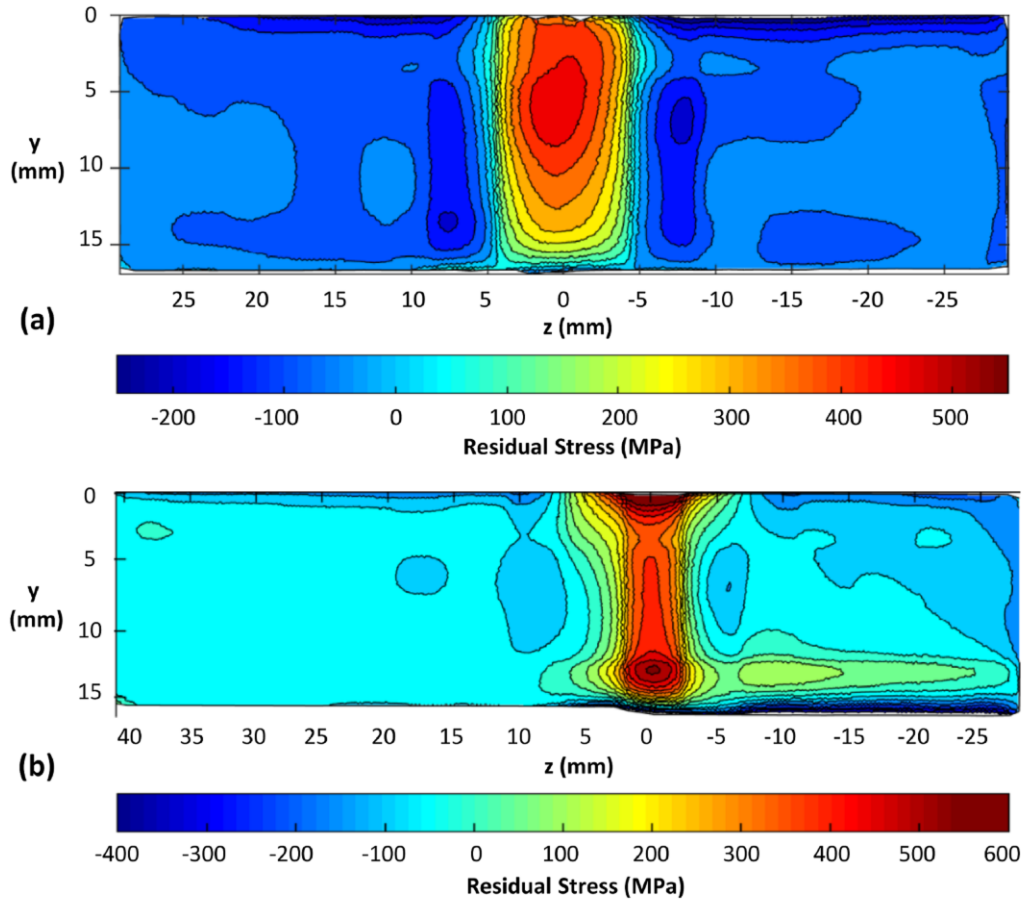


**Fig. 9.** Average surface height data measured by CMM for (a) Ring 1, and (b) Ring 2.

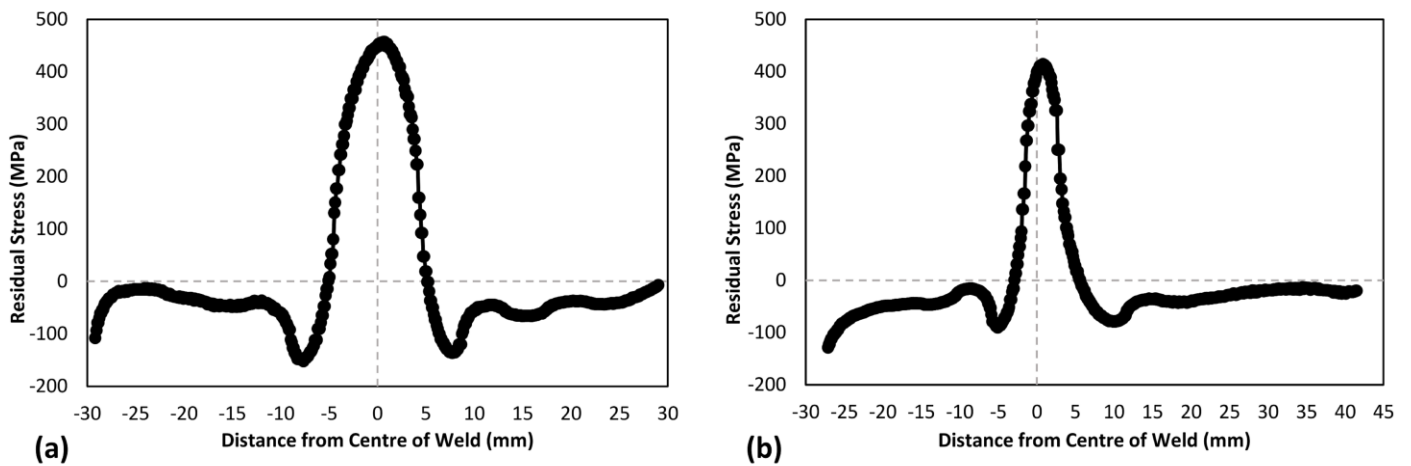
Considering the 2D hoop residual stress maps for both rings, it is clear that residual stress in the weld region decreases almost linearly in the  $-z$  and  $+z$  direction, with Ring 1 and 2 displaying differing linear and hyperbolic shapes respectively. Such a relationship agrees with the microhardness data in Fig. 6 where Ring 1 exhibits a wider region of high hardness over the weld zone compared to Ring 2. This is shown in Fig. 11 by plotting the residual stress profiles across the whole weld cross-section along the  $z$ -axis. Additionally, from Fig. 10b it was noted that Ring 2 exhibited a protrusion in tensile residual stress along the inner diameter of the ring, up to roughly 27 mm away from the weld in the  $-z$  direction. A corresponding balancing zone of highly compressive residual stress was observed towards the ring inner surface, away from the tensile protrusion, with a maximum value of  $-625$  MPa.

Plots of surface, near-surface, and through-depth residual stress were produced by combining the results of all investigated measurement techniques for each welded ring. Particular focus was given to the FZ in the weld centre (i.e.  $x = z = 0$  mm) and the BM at a distance of  $\approx +20$  mm from the weld centre ( $x = 0$  and  $z = +20$  mm) on the outer surface. The residual stress profiles measured by contour method, extracted from the 2D contour plot data presented in Fig. 10, and the results obtained by XRD and hole-drilling based on ESPI are presented as functions of depth within 2.5 mm of the outer surface. These results are shown in Fig. 12. Contour method data was excluded for depths between the surface and 0.5 mm due to the reduced accuracy of the technique within this range when compared to that of the other employed techniques [23].

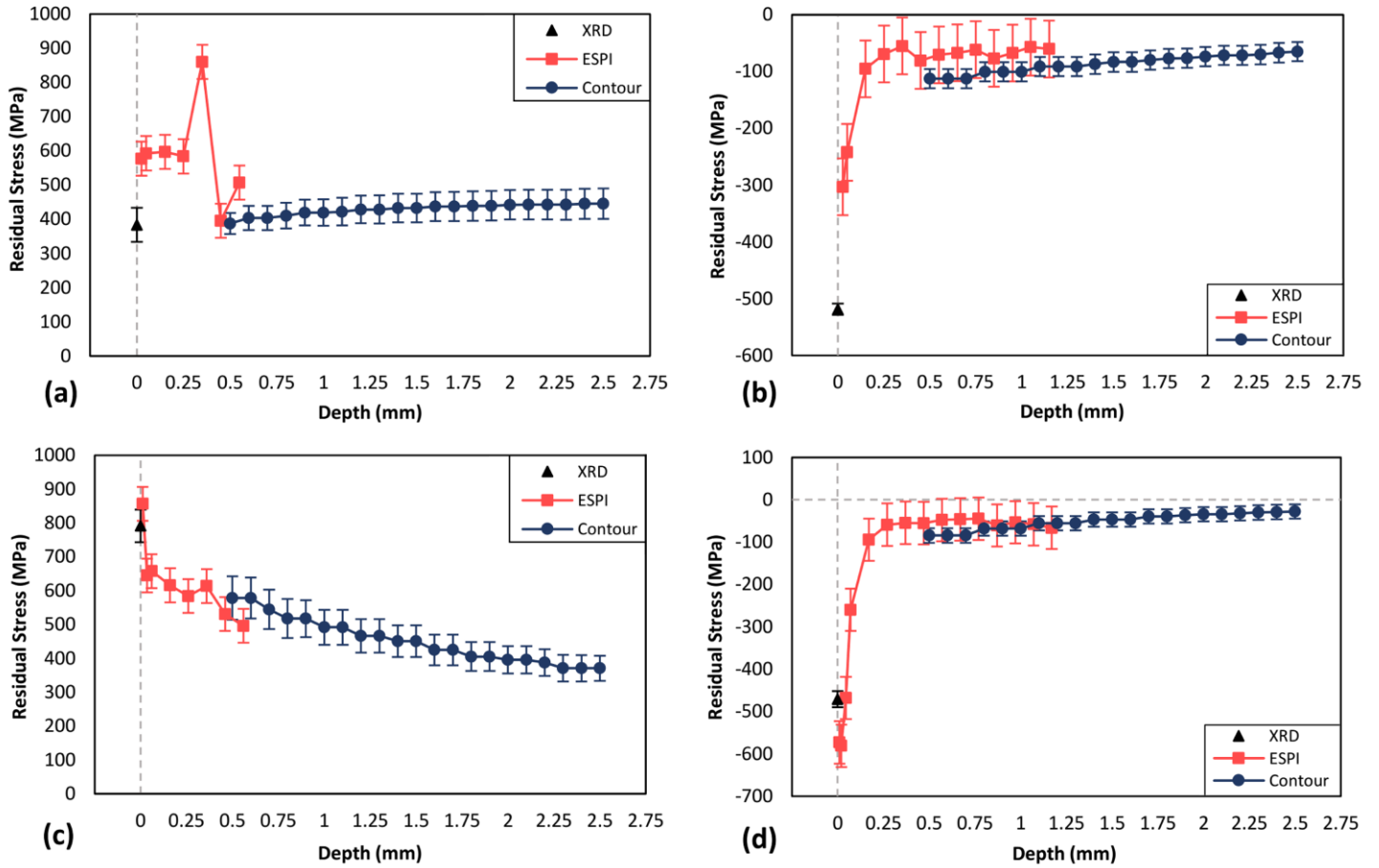




**Fig. 10.** Out-of-plane 2D hoop residual stress maps of samples as determined by the contour method and bending moment for (a) Ring 1, and (b) Ring 2.



**Fig. 11.** Residual stress profiles across the weld cross-section (i.e. along the  $z$ -axis in Fig. 10) taken from the residual stress maps measured by contour method presented in Fig. 10 for (a) Ring 1 at  $y = 8.4$  mm, and (b) Ring 2 at  $y = 8.2$  mm.



**Fig. 12.** Combined results of XRD, hole-drilling based on ESPI and contour method hoop residual stress as functions of depth ( $y$ ) within 2.5 mm of the outer surface for (a) FZ of Ring 1 at  $z = 0$  mm, (b) BM of Ring 1 at  $z = +20$  mm, (c) FZ of Ring 2 at  $z = 0$  mm, and (d) BM of Ring 2 at  $z = +20$  mm.

The plot of surface and near-surface depth stress profile of the FZ of Ring 1 (see Fig. 12a) displays tensile data for all applied techniques. Upon consideration of the depth plot in the BM of Ring 1 (see Fig. 12b), the residual stress distribution is shown to be compressive for all points. Similarly, all data points for the FZ (see Fig. 12c) and BM (see Fig. 12d) of Ring 2 are tensile and compressive, respectively. There is good agreement between contour results and ESPI based hole-drilling data in the FZ and BM of both rings. This is especially evident in the BM of Ring 1 and Ring 2 from depths of 0.5 mm to approximately 1.15 mm, with all contour and ESPI data within the bounds of measurement error throughout this range [23].

#### 4. Discussion

The residual stress profiles obtained using XRD, hole-drilling with ESPI and the contour method showed good agreement, verifying the results of the contour method and allowing comparison with microstructural observations. Since the cut for contour method was conducted after the relaxation of bending moments, the errors associated with plasticity in the tip of the cuts were minimised [23]. The contour method provided an approach to easily measure and evaluate through thickness hoop residual stress. However, the technique has some limitations with regards to sensitivity. High residual stress gradients were present in the parts; identified across the weld and near the BM sample surface. These stress distributions may not be accurately depicted due

to the selected element size being too large and/or the selected spline knot spacing over-smoothed the surface topography data, and in turn affecting the residual stress results [28]. Knot spacing and element size selection also leads to extracted contour data points representing a number of small segments with averaged residual stress in place of point measurements (e.g. XRD or ESPI based hole-drilling) which can better capture peaks in residual stress [23]. For the purposes of understanding bulk residual stress distributions across the weld, the wavelength covered by the contour method is adequate. However, it is also often important to have an understanding of surface and near-surface residual stresses with high spatial sensitivity due to the role of such residual stresses during fatigue crack initiation [7]. Therefore, the utilisation of XRD and ESPI based hole-drilling proved complimentary to the contour method by offering a means of contour verification as well as determination of high fidelity surface and near-surface residual stress with relatively short wavelength which cannot be captured accurately by contour method.

The general 2D through thickness distribution of hoop residual stresses in both rings, as presented by the contour method, depicted a tensile region in the FZ. This decreased in magnitude across the HAZ, developing into highly compressive pockets in the BM, and then slightly decreasing in magnitude to a relatively steady compressive value as distance from weld increased. This distribution is as would be expected for a typical electron-beam weld [3, 15].

The peak tensile hoop residual stresses of both rings were obtained by ESPI based hole-drilling in the FZ. These were recorded at similar depths;  $+ 860 \pm 50$  MPa at a depth of 0.35 mm for Ring 1 and  $+ 898 \pm 50$  MPa at a depth of 0.36 mm for Ring 2. This is similar to the maximum residual stress of  $+ 914$  MPa measured by Xie *et al.* using the contour method on electron-beam welded Ti-6Al-4V plates of 50 mm thickness [15]. In order to compare the residual stress to the material yield strength, the effect of varying microstructures across the weld must be considered. Wang and Wu linked the microstructure to the yield strength of electron-beam welded Ti-6Al-4V joints, with the BM, HAZ and FZ corresponding to yield strengths of 932 MPa, 892 MPa and 857 MPa respectively [18]. There are differences between the aforementioned study and the work conducted in terms of material geometry, weld conditions and BM microstructure. However, based on these properties the peak tensile stress recorded in the FZ would be in the range of the material yield strength. Such highly tensile residual stress is undesirable and post weld heat treatment may be applied to reduce such residual stresses in industry [3, 15].

Although the obtained data provides evidence to support the expected general trends in residual stress distribution, there are some dissimilarities in magnitudes and profiles between the two investigated rings. These dissimilarities may be due to the differences in welding conditions, sample geometry and additional cosmetic pass. For example, the tensile region in Ring 2 is smaller than that in Ring 1, which can be attributed to the reduced welding power used. This leads to a lower heat input, reducing the size of the FZ and in turn, the size of the tensile zone [3, 29]. Additionally, the axial residual stresses measured by XRD and ESPI based hole-drilling showed a tensile region in the FZ and HAZ on the outer diameter of Ring 2; whereas Ring 1 exhibited compressive axial residual stresses at all surface and near-surface measurement points. This difference is thought to be attributed to the additional cosmetic pass on the outer diameter of Ring 2.



The residual stress distribution in the FZ of both welded rings was found to be highly tensile throughout. This tensile zone was associated with a peak in microhardness and the observation of an almost entirely acicular martensitic structure in the FZ. Hardness within Ti-6Al-4V welds is typically directly related to the volume fraction of  $\alpha'$ , hence the highest values were recorded in the FZ [17]. This microstructural observation agrees with the work of Gao *et al.* [30] who confirmed that unlike TIG welded Ti-6Al-4V plates, the FZ in a Nd:YAG laser weld was formed of pure  $\alpha'$  rather than  $\alpha + \alpha'$ . Additionally the observed isolated peaks and troughs in microhardness within the bulk are likely product of differing alpha/beta spacing, causing variations in boundary hardening, at those measurement points [20].

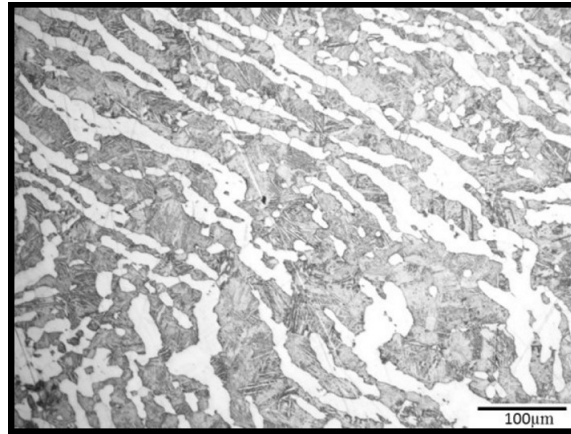
During welding, the local temperature in the FZ is sufficient to cause melting, which upon re-solidification results in a displacive martensitic transformation ( $\beta \rightarrow \alpha'$ ) due to the surrounding bulk material and atmosphere acting as heat sink. The temperature gradient between weld and surface/bulk material causes the formation of thermal stress, which upon high cooling rates leads to the development of internal tensile residual stress, with surrounding regions of balancing compressive residual stress into the BM [20]. Thus, the temperature transition during martensitic phase transformation is likely to be the main source of tensile residual stress generation in the FZ. Additionally, the phase transformation itself may play some role in the evolution of residual stress during welding [24]. The  $\beta$  (BCC)  $\rightarrow \alpha'$  (HCP) transformation of titanium alloys is a shear transformation like that of other martensitic transformations in other materials (e.g. steels) and is therefore characterised by an increase in stored strain energy in order to accommodate an associated volume change. Thus, the observed martensitic microstructure in the FZ can be linked to the presence of tensile residual stresses in this region of the electron-beam welded components.

The maximum tensile residual stress measured by the contour method, + 755 MPa, was found to be located within a secondary HAZ identified near the outer surface of Ring 2 in the FZ. This is attributed to the secondary cosmetic pass applied to the ring, and coincides with the observation of ghost  $\alpha$ . Additionally, as shown by Owen *et al.* [31], an increase in ghost  $\alpha$  volume fraction can lead to an increased hardness compared with  $\alpha'$ . This correlates well with the observed increase in microhardness results over the secondary HAZ, providing further evidence to support the observation of ghost  $\alpha$ . As ghost  $\alpha$  is rare in comparison with  $\alpha$ ,  $\beta$  and  $\alpha'$  phases, there is a relative lack of literature surrounding it. Nevertheless it has been observed in a number of welding techniques, both solid state and fusion based [32]. Attallah *et al.* [33] reported that under sufficiently high cooling rates, when the local temperature approaches the beta transus,  $\beta_s$ , there is insufficient time to permit the diffusion of alloying elements leading to their segregation in the transformed  $\beta$ , the product of which leads to the formation of lamellar ghost  $\alpha$ , rich in aluminium, with a semi-continuous structure and similar appearance to  $\alpha_{gb}$ . Le Biavant *et al.* [34] assessed the crystallographic orientation of ghost  $\alpha$  in forged Ti-6Al-4V, determining a strong texture across macrozones. This texture was two orders of magnitude greater than the surrounding grains. The additive effect of this strong texture coupled with a highly tensile residual stress in this region will lead to accelerated fatigue crack initiation, reducing fatigue performance of the part [34].

The HAZ micrographs of both welded rings showed evidence of a transitional zone from the martensitic microstructure of the FZ to the characteristic bimodal  $\alpha + \beta$  microstructure of the BM. This coincided with a

reduction in tensile residual stress and microhardness levels as distance from FZ increased. The decrease in residual stress can be attributed to the reduction in the percentage of  $\alpha'$  across the HAZ. However, there was a further reduction in residual stress past the point that the bulk microstructure was reached, with troughs of compressive residual stress noted beyond the microstructurally defined HAZ of both rings. The observation of similar highly martensitic microstructures yet differing phase morphologies in the near HAZ and FZ can be explained by the results of Greenfield and Duval [32], who state that the phase morphology of the near HAZ and FZ is significantly distinct due to increased solute segregation during solidification in the FZ.

The unexpected protrusion of tensile residual stress observed in the BM of Ring 2 was further investigated by carrying out optical microscopy of a material sample from the identified region. A micrograph of this sample is provided in Fig. 13, showing the presence of elongated  $\alpha_{gb}$  grains, over 400  $\mu\text{m}$  in some cases, within the protrusion zone. The protrusion reaches far beyond the HAZ and is therefore not directly caused by the welding process and the microstructural changes induced by welding. Instead, it thought to be geometrically necessary stresses to balance the significantly high compressive residual stress on the inner surface induced by a slight difference in the thicknesses of the BM used in either side of the weld. Or, it may well be a product of the thermomechanical history during production of the base material prior to the EBM welding process. Significant effort is applied during manufacturing to ensure the breakup of these structures in an attempt to avoid deleterious effects on crack initiation and fatigue [35]. As these microstructural features coincide with the residual stress protrusion, it is highly probable that the two findings are interrelated. This further highlights the potential use of the contour method for the identification of problem areas in a manufactured part.



**Fig. 13.** Optical micrograph of elongated  $\alpha$  extending from the primary HAZ of Ring 2.

## 5. Conclusions

Microstructural observations and microhardness measurements were successfully linked to the through-thickness residual stress distribution in Ti-6Al-4V electron-beam welds, as measured using a combination of the contour method, XRD and hole-drilling with ESPI. It was found that:

- The FZ for both electron-beam welded rings was characterised by a highly tensile region of residual stress, corresponding to a microstructure which primarily consisted of acicular  $\alpha'$  in prior  $\beta$  grains; formed upon solidification.

- A decrease in tensile residual stress from near to far HAZ was observed in both welded rings. This was also marked by a decrease in the volume fraction of acicular  $\alpha'$  and microhardness, as well as a shift towards the bulk microstructure.
- The BM was found to principally exhibit compressive residual stress, and contained a bimodal microstructure of elongated  $\alpha$  and  $\alpha$  lamellae in a matrix of transformed  $\beta$ .
- The transition from tensile to compressive residual stress upon increasing distance from weld centre correlates with the shift from a shear or displacive dominated transformation to a diffusional dominated transformation in the HAZ.
- A secondary HAZ was identified near the outer diameter of the FZ of Ring 2 as a result of the application of a secondary cosmetic pass weld. This region possessed microstructural evidence of 'ghost  $\alpha'$ ' formation in combination with exceptionally high levels of tensile residual stress.
- Additionally, the contour method was used to identify an area of unexpected tensile residual stress in the BM which, upon further investigation, had microstructural dissimilarities compared to the remainder of the BM.

These results may be used to advance the development of predictive models for weld behaviour, allowing for the optimisation of microstructure and residual stress in electron-beam welded Ti-6Al-4V components.

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